

# X-rays and Neutrons for Directed and Self-Assembly

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X-rays and

# Neutrons for Directed and Self-Assembly

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# Overview

- Self-Assembly Characterization
- Chemical Transformation - Polymerization
- External Field - Rheology and Flow Alignment
- SANS Method
  - Isotope variation (H and D) to provide contrast

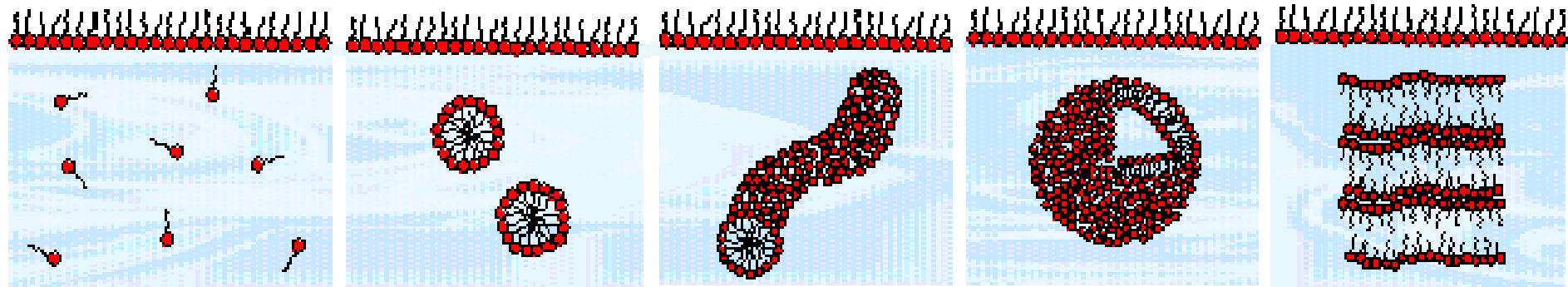
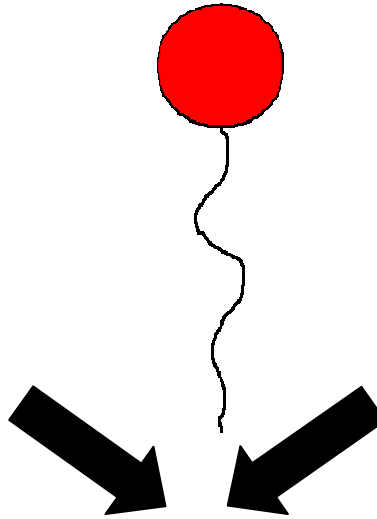
# Aqueous Surfactants

## Surfactant structure

- Hydrophilic Head Group
- Hydrophobic Tail
- Counterion

## Solution Conditions

- Temperature
- Concentration
- Salts
- pH
- Additional Molecules
  - ex. Hydrotropes



Monomers

Micelles

Elongated Micelles

Vesicle

Lamellar

## Macroscopic Properties

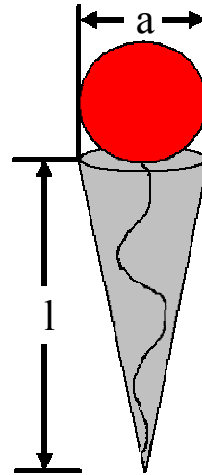
- Surface Tension
- Viscosity
- Solubility

# Aqueous Surfactants

## Packing Parameter

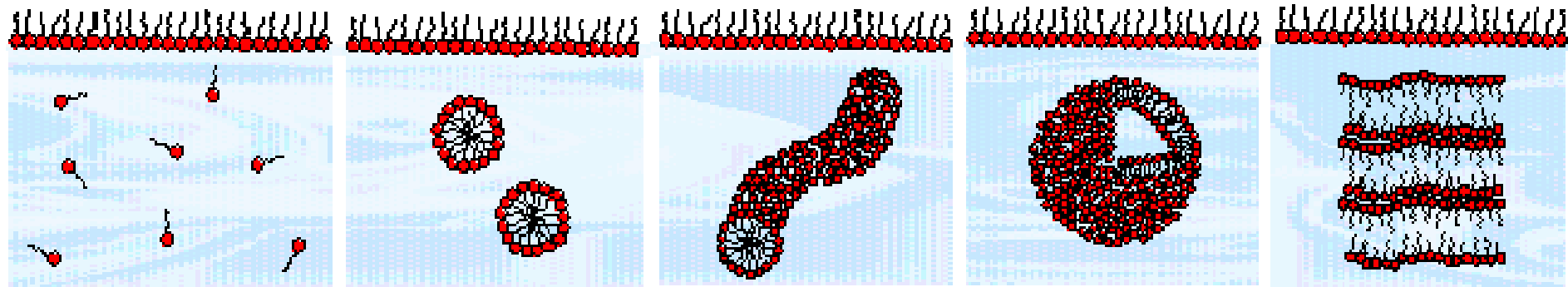
$$P = \frac{v}{al}$$

- $a$  – Head group cross-sectional area
- $l$  – Length of tail
- $v$  – Volume of tail



## Solution Conditions

- Temperature
- Concentration
- Salts
- pH
- Additional Molecules  
– ex. Hydrotropes



Monomers

Micelles

Elongated Micelles

Vesicle

Lamellar

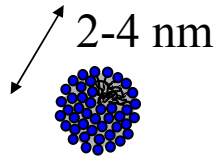
**P**  $\longrightarrow$   $\leq \frac{1}{3}$   $\longrightarrow$   $\leq \frac{1}{2}$   $\longrightarrow$   $\leq 1$   $\longrightarrow$   $\approx 1$

Macroscopic  
Properties

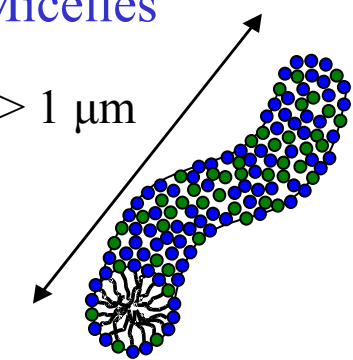
- Surface Tension
- Viscosity
- Solubility

# Self-organized surfactant solutions: promising tool for the templated (directed) synthesis of organic and inorganic materials

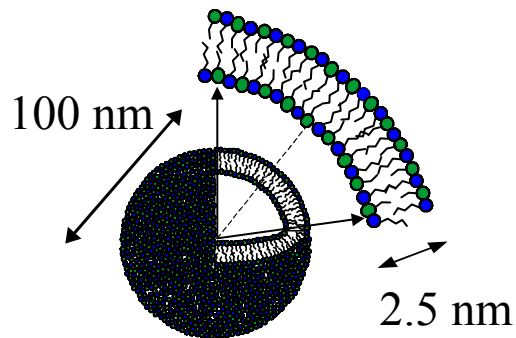
Spherical  
Micelles



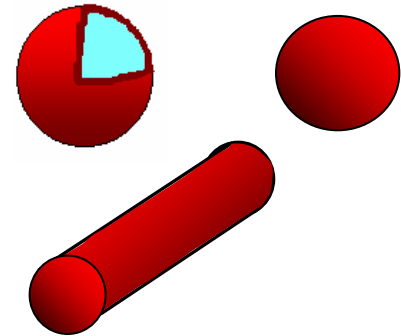
Rod-like  
Micelles



Vesicles

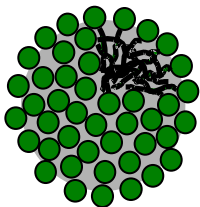


Polymerization

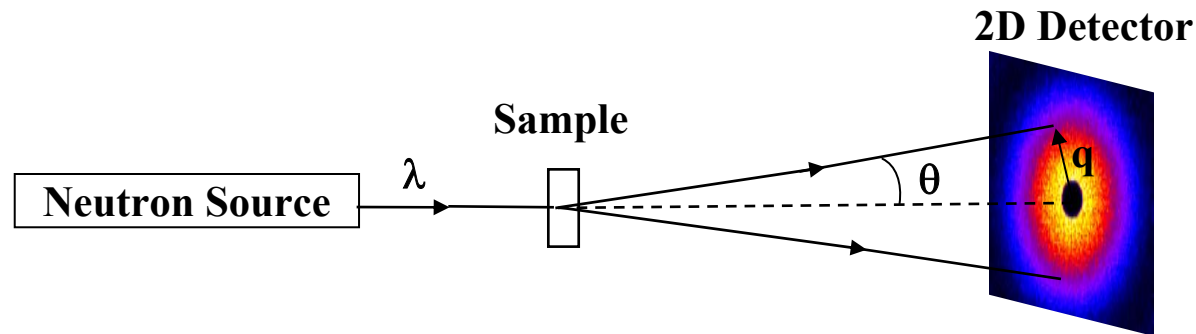


- Increased strength
- Stability to solution conditions
- Capture aggregate dimensions

Microemulsion Droplets



# Small Angle Neutron Scattering (SANS)



$$q = \frac{4\pi}{\lambda} \sin\left[\frac{\theta}{2}\right]$$

$$q \sim \frac{1}{\text{Length}}$$

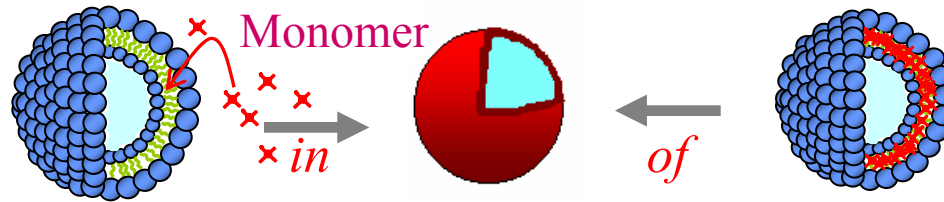
$$I(q) = n P(q) S(q)$$

$P(q) \Rightarrow$  Form Factor  $\Rightarrow$  Single particle properties (size, shape, composition)

$S(q) \Rightarrow$  Structure Factor  $\Rightarrow$  Relative positions of particles due to interactions



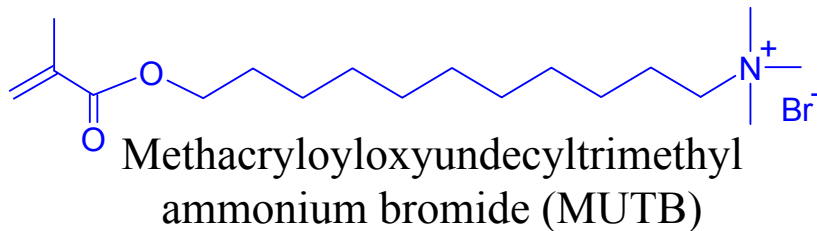
# Vesicle Templating



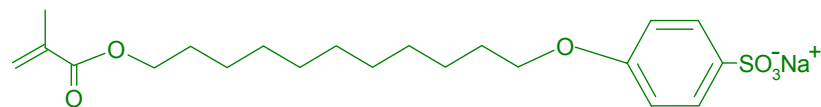
McKelvey et al. Langmuir, 2000

Liu et al. Langmuir, 2003

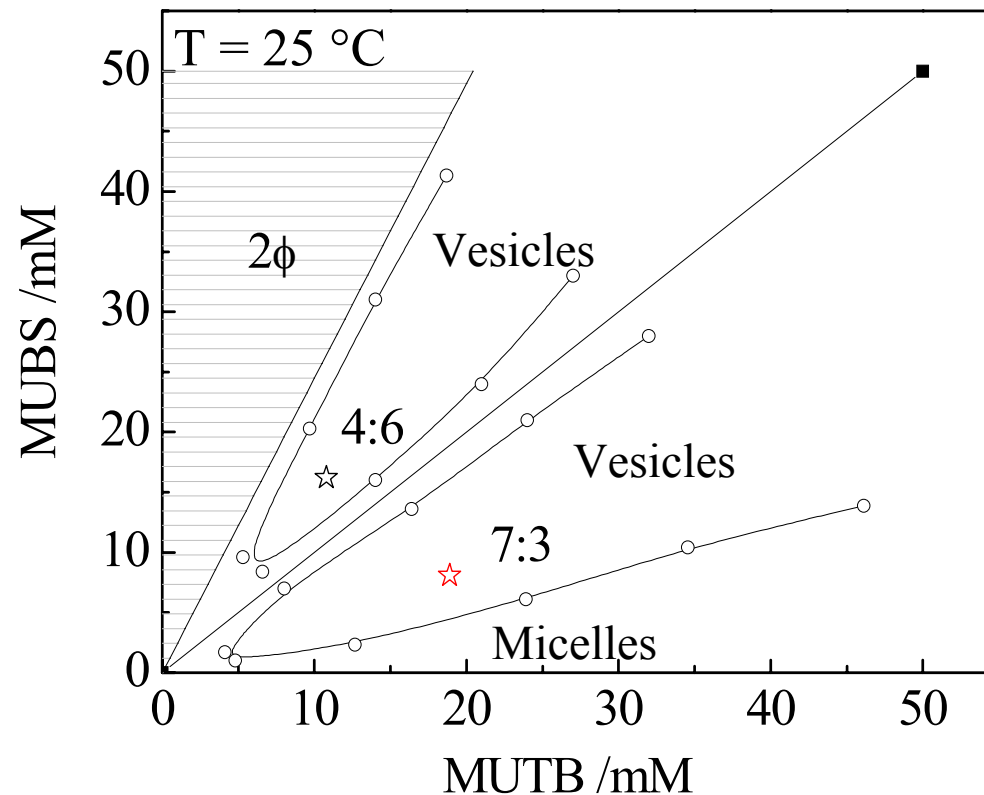
- Polymerization *in* or *of* vesicles allows replication of their architecture
- Fixation of vesicles (*of*):



Polymerizable surfactants



Sodium 4-ω-methacryloyloxyundecyl) oxybenzene sulfonate (MUBS)

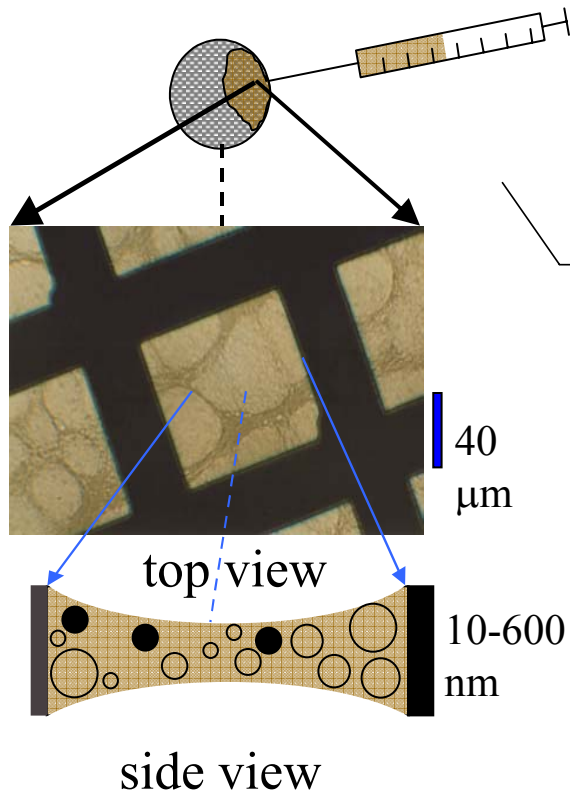




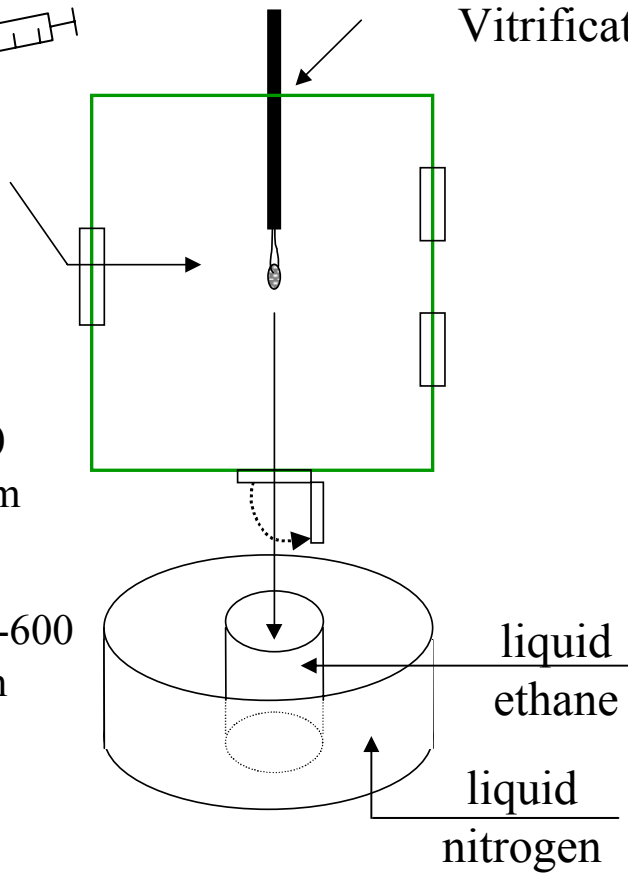
# Cryogenic Transmission Microscopy (cryo-TEM)

Sample preparation goal: lock in microstructure by rapid cooling

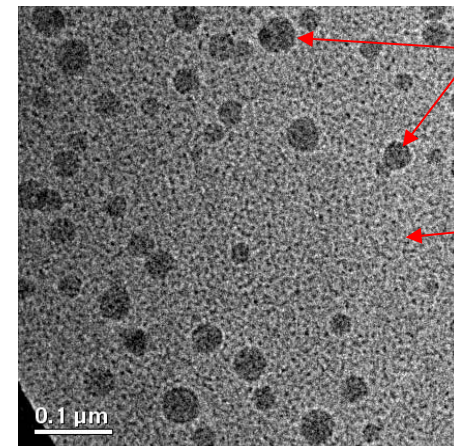
1. holey carbon grid



2. CEVS  $\Rightarrow$  Controlled Environment Vitrification System

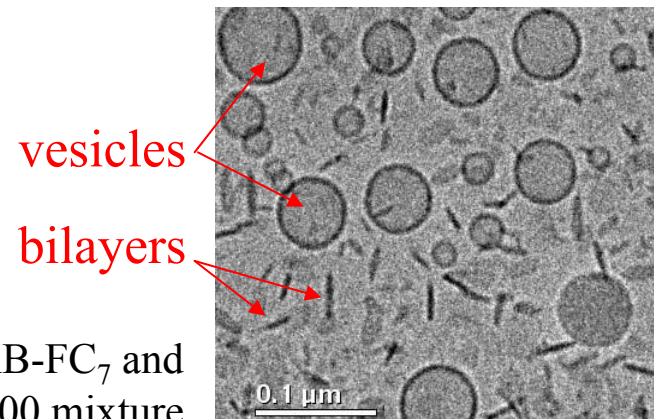


3. TEM imaging: 100 keV



beads of p-C6MA

micelles of DTAB & DDA

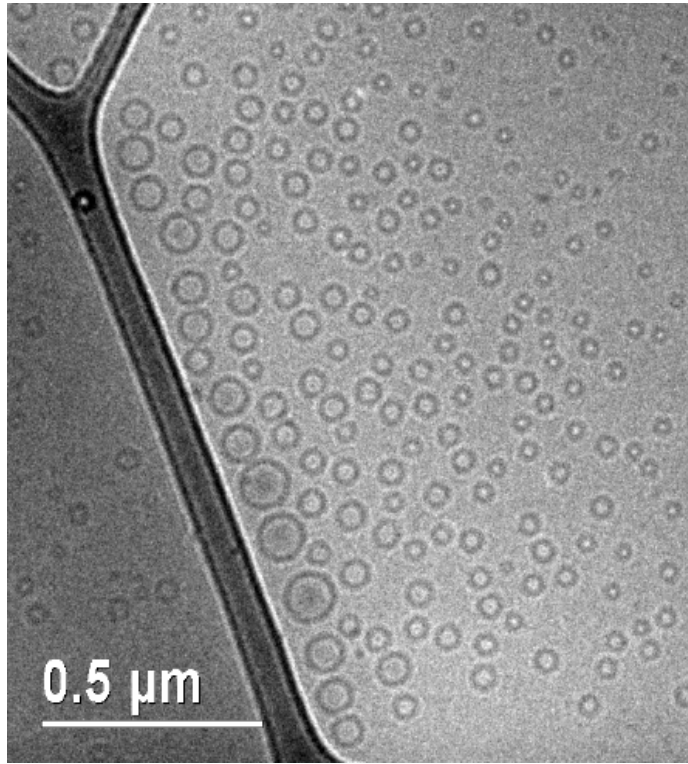


vesicles

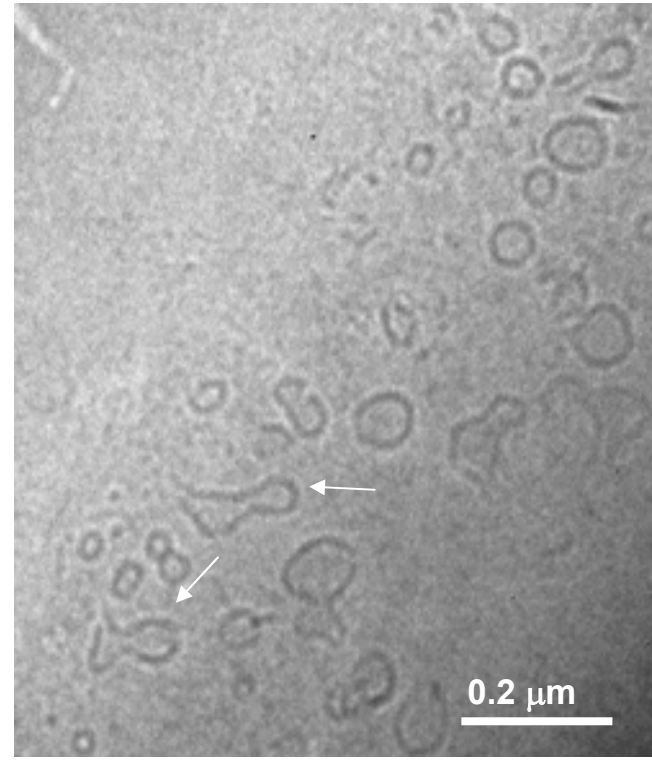
bilayers

CTAB-FC<sub>7</sub> and IR-400 mixture

# Cryo-TEM of Polymerizable Vesicles



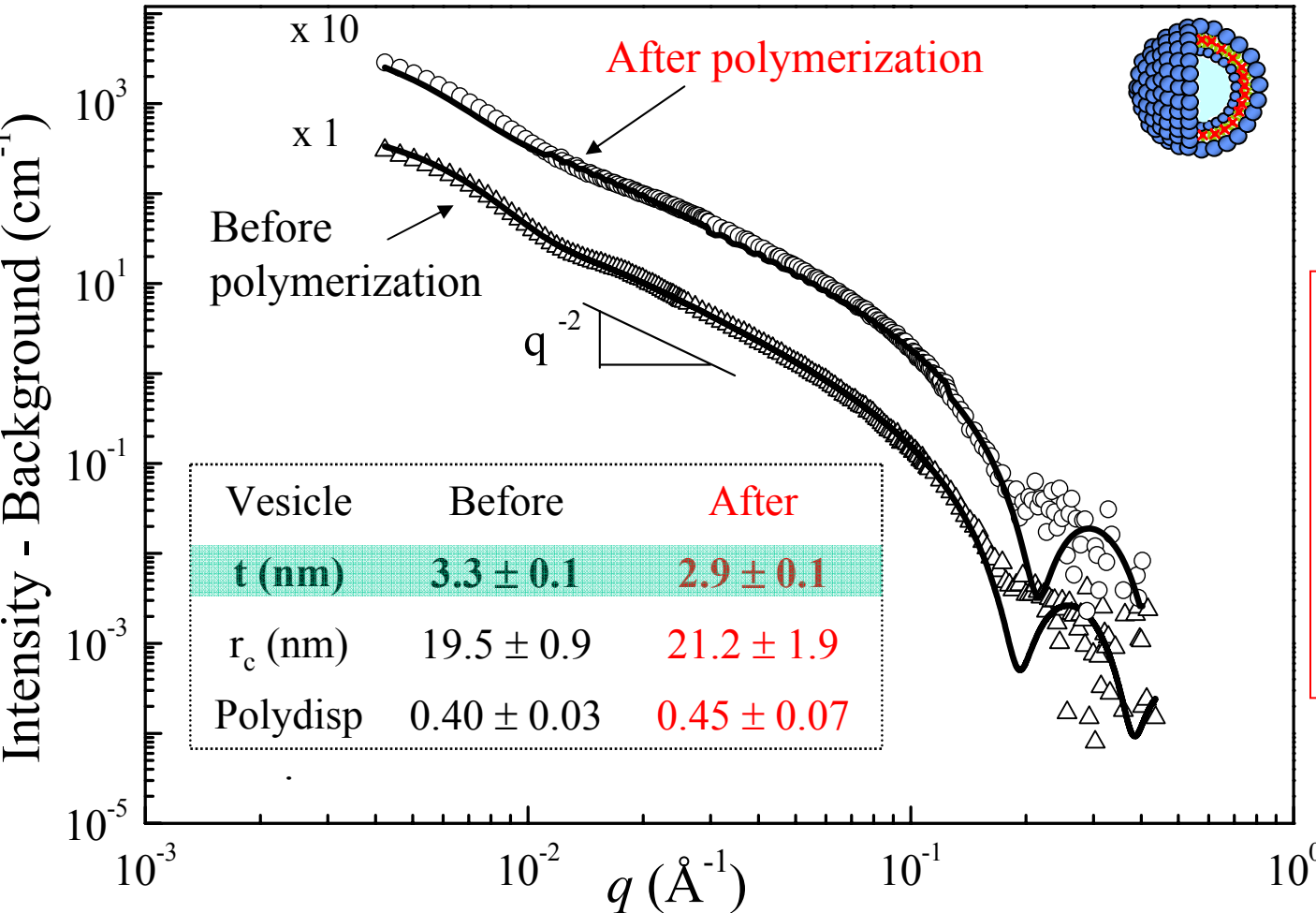
Before polymerization; QLS = 98 nm



After polymerization; QLS = 100 nm

- Size of the polymerized vesicles is comparable to that of the original vesicles
- A mixed population of spherical and irregular shaped unilamellar vesicles
- Complex morphology could be caused from the use of ionic initiator or from the reaction process

# SANS Spectra of Polymerizable Vesicles



$$I(q) = n P(q) S(q)$$

Model of  $P(q)$ :

- Polydisperse-core-shell
- Schulz distribution
- Minimized  $\chi^2$

- Bilayer thickness reduction due to polymerization of double bonds
- Polymerized vesicles appear to be a replica of the original vesicles

# SANS Characterization

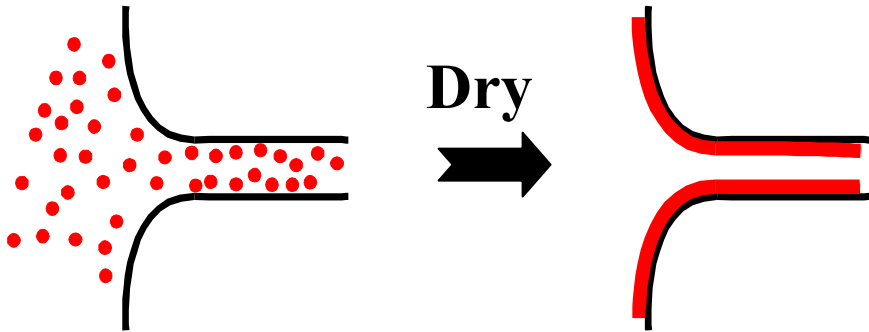
- Determination of nano-scale self assembly dimensions and interactions
- Polymerization can capture structures
- Contrast variation can isolate properties

# Why Study Microemulsion Polymerization?

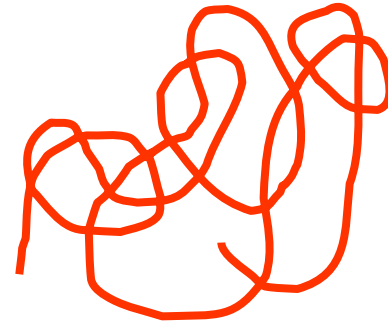
Produces nanosized ( $\sim 15$  nm) latex particles smaller than those obtainable by emulsion polymerization

Polymerization in a confined environment may lead to unique polymer morphologies, e.g. tacticity and knotting.

**“Paint” the walls of a microporous material**



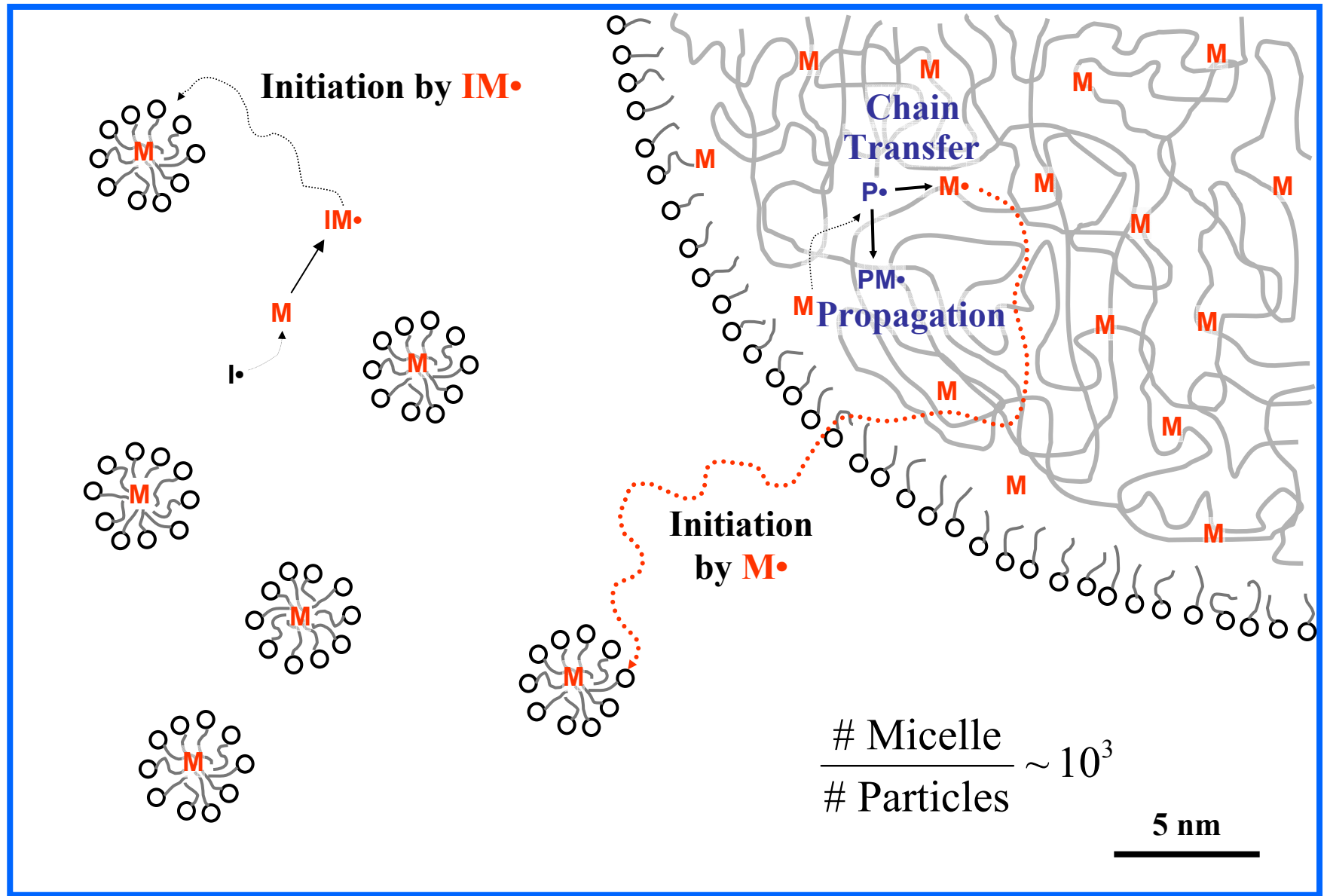
**“Knotted” polymer chain in solution**



**“Seeds” for emulsion polymerization**

**Extremely high MW  
( $\sim 20$  million daltons)  
are readily made**

# A Simple View Of Microemulsion Polymerization

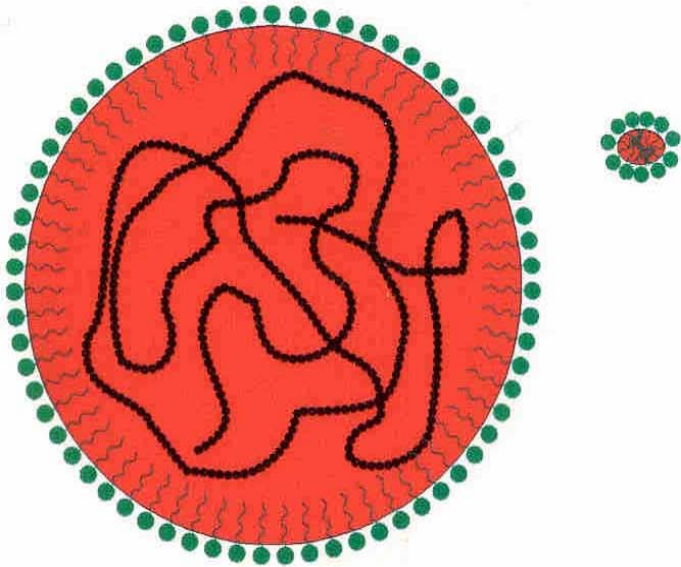




# Kinetic Modeling: Monomer Concentration at Locus of Polymerization

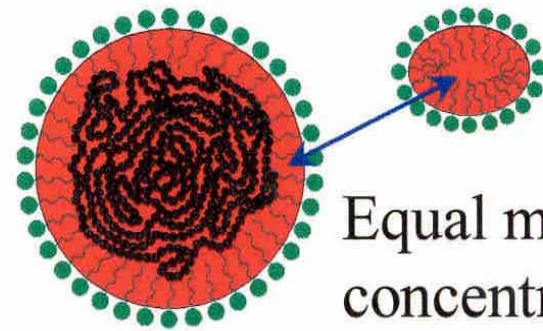
At intermediate concentrations, which is closer to the truth?

## Monomer Swells Polymer



$$C_p \approx C_{\max}(1 - f)$$

## Monomer Does Not Swell Polymer



Equal monomer concentration

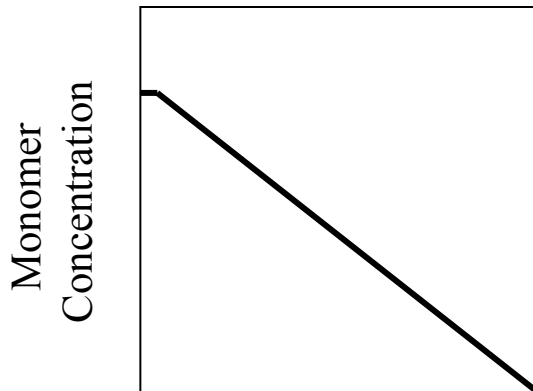
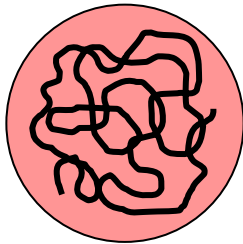
$$C_p \approx C_o(1 - f)$$

$C_o \Rightarrow$  Initial micelle core monomer concentration

# How Does the Microstructure Evolve?

## Case I

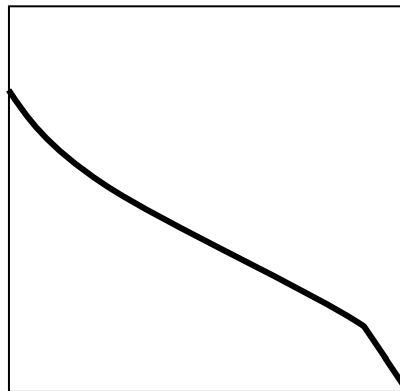
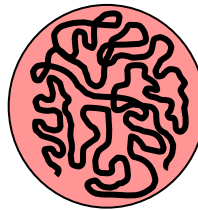
Flory-Huggins for both latex particles and micelles.  
All monomer is taken up by latex particles at  $\sim 5\%$  conversion.



Conversion

## Case II

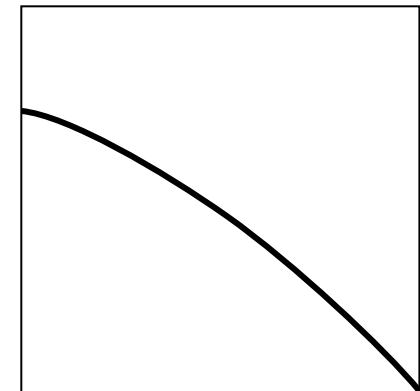
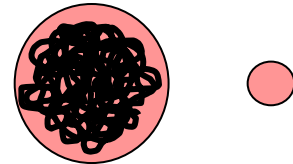
Flory-Huggins for latex particles and curvature energy for micelles.  
Monomer partitions between latex and micelles.



Conversion

## Case III

No swelling of polymer latex. Polymerization occurs in shell of latex particles with monomer concentration equal to that in the micelles.

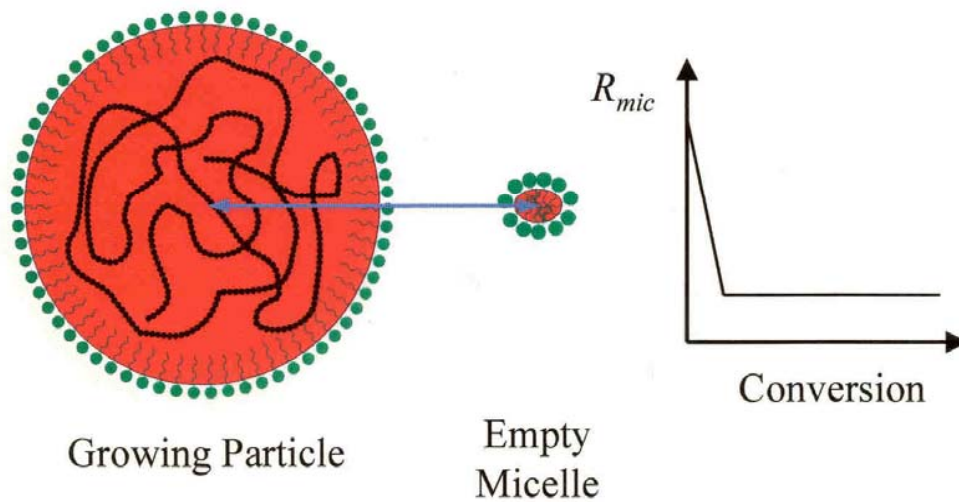


Conversion

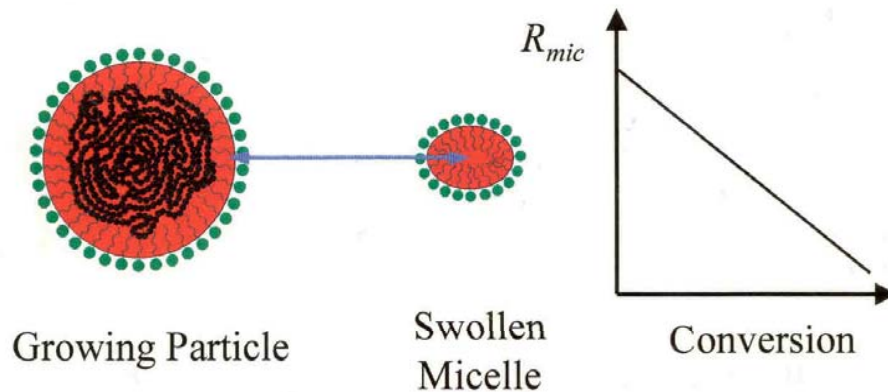
**Monomer concentration is approximately linear with conversion.  
Can differentiate only using Small Angle Neutron Scattering (SANS).**



## A) Swollen Chain



## B) Compact Chain

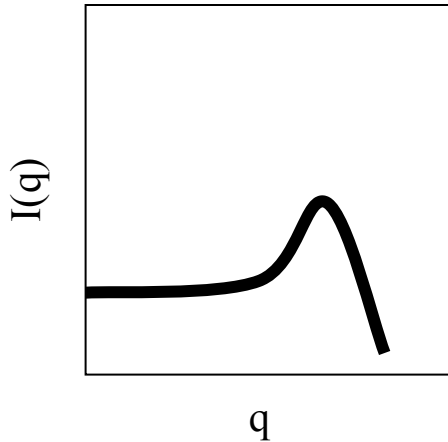


Monomer Concentration at  
Locus of Polymerization:

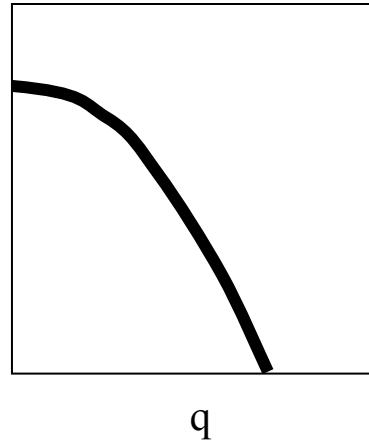
$$c = c_o(1-f)$$

# What Can SANS Tell Us?

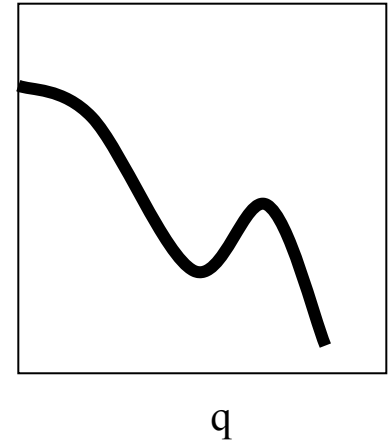
Micelle  
Scattering



Particle  
Scattering

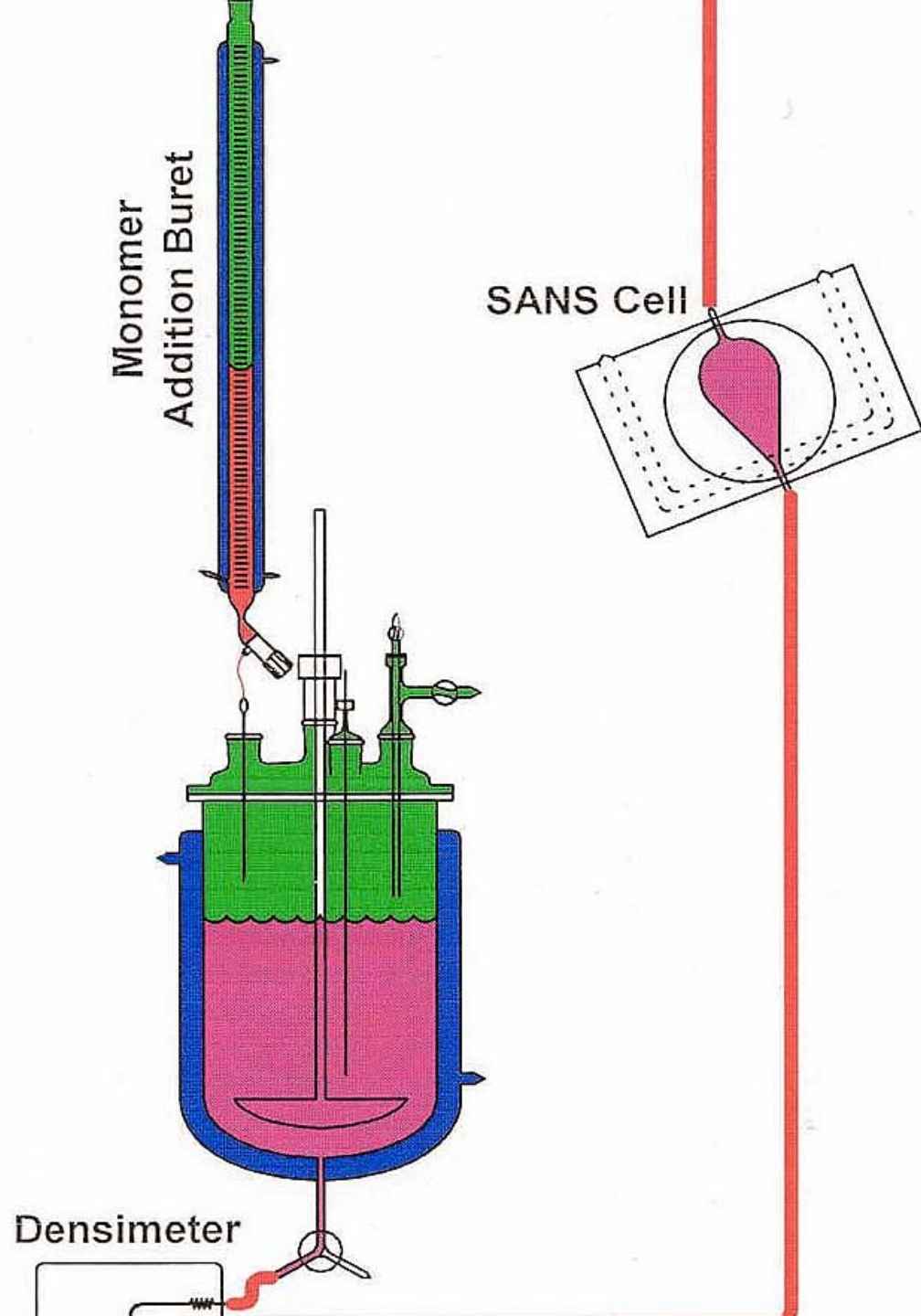


Observed  
Scattering



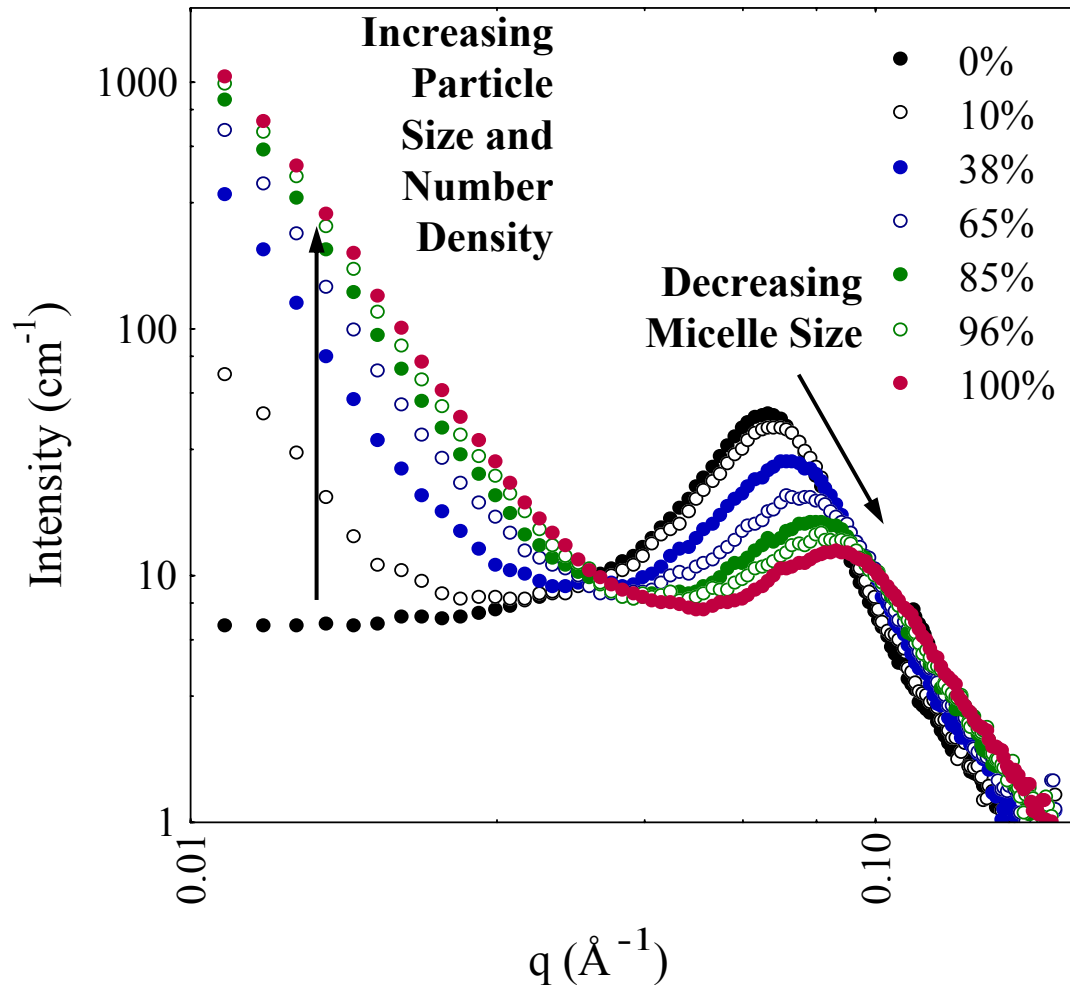
**How does the SANS spectra change as the microstructure evolves from micelles to a mixture of polymer particles and micelles?**

# Reactor for Online Scattering During Polymerization



# Online SANS / Kinetics Experiments

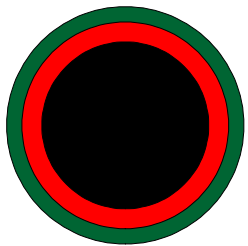
C6MA (DTAB/DDAB)



**Gradual shifts in SANS spectra indicate that monomer partitions between the micelles and the polymer particles. Case I is incorrect.**

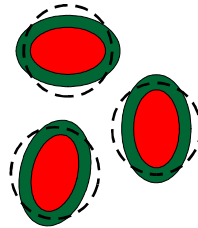
# SANS Model for Online SANS Spectra

Form Factors

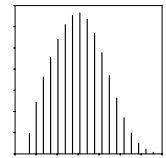
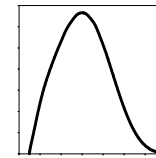


Polymer

Effective HS Interactions

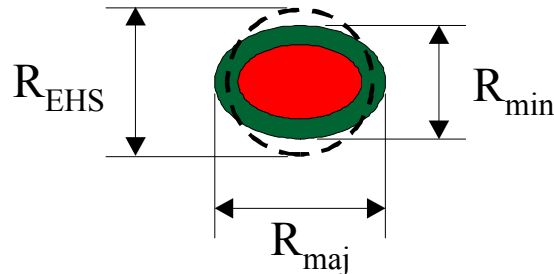


Discretize Model-Predicted  
Particle Size Distribution

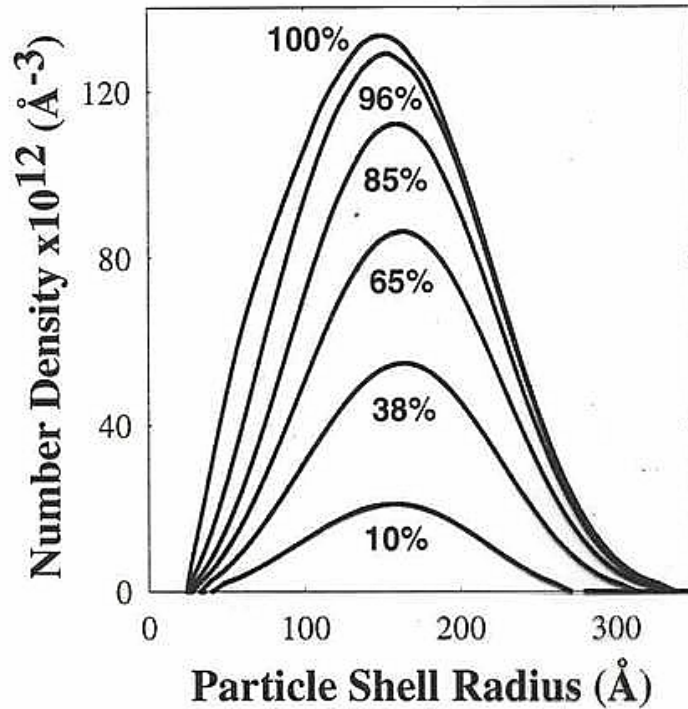


Calculate model intensities using Vrij's analytical equation

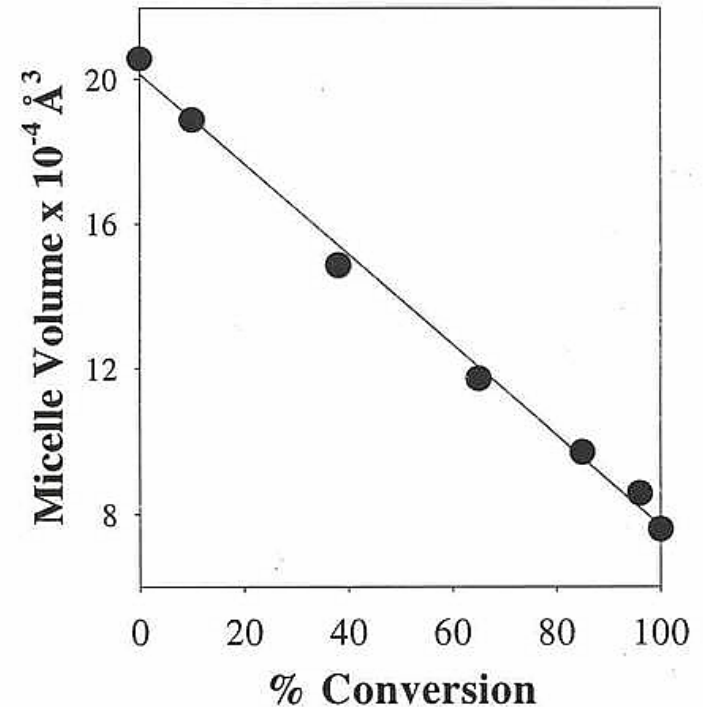
Three Adjustable  
Parameters



# SANS Model Fitting Results



**Particle size distribution  
model is consistent with SANS**



**Micelle size decreases steadily  
with increasing conversion**

# Summary

- Free radical polymerization in and of complex fluids produces replicas
- SANS can record the dynamics of chemical transformations
- Faster physical transformations can be observed with stopped flow or other methods

# Rheology and SANS

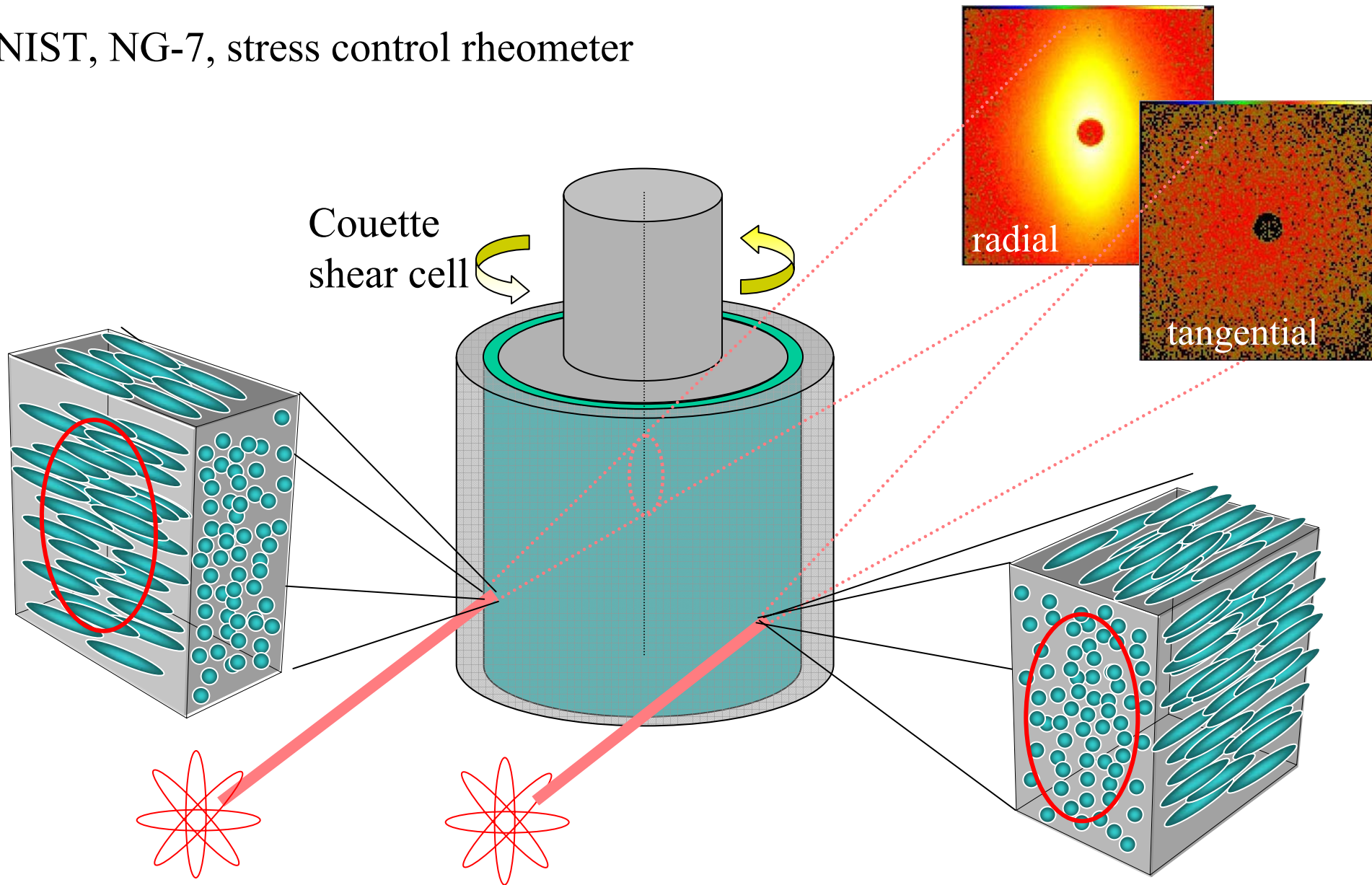
- Structural information on complex fluids in the size range 1nm~1 micron both at rest and under **flow** conditions
- Access to **rheological** properties not accessible in a mechanical rheometer
- Contrast matching in SANS enables probing individual **components** in a mixture
- Initial target – ordering in worm-like micellar solutions

With Norm Wagner – University of Delaware

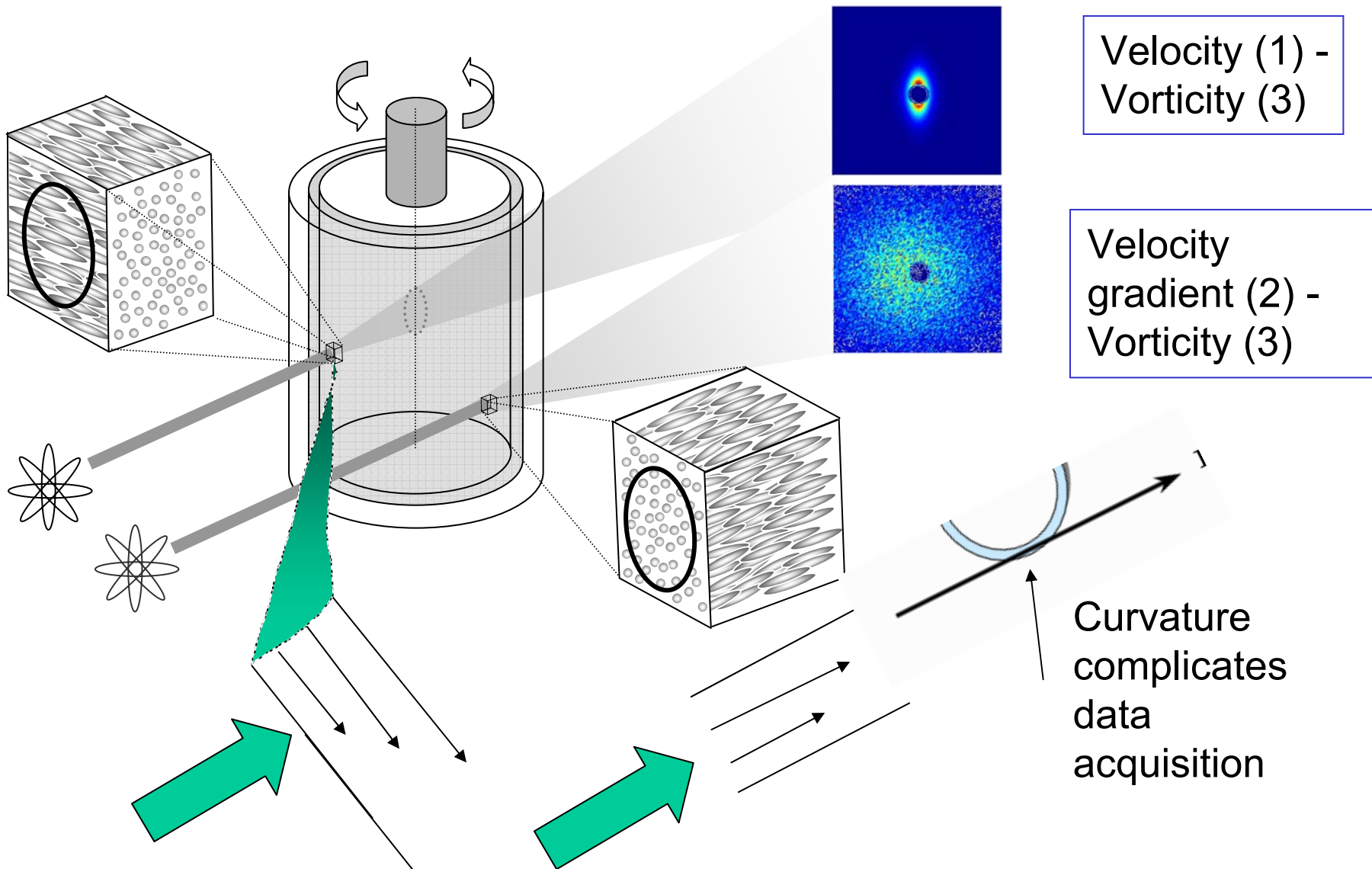


# Rheo-SANS Investigation

NIST, NG-7, stress control rheometer



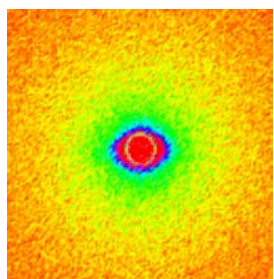
# Current Rheo-SANS Geometries



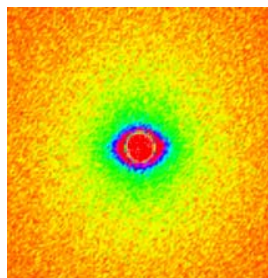
# Rheo-SANS Couette Cell



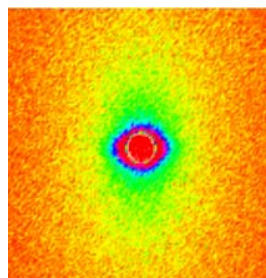
# Rheo-SANS Comparison



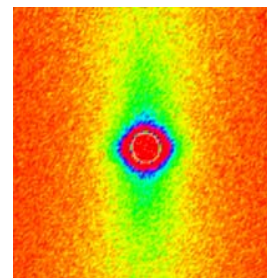
0.1 s<sup>-1</sup>



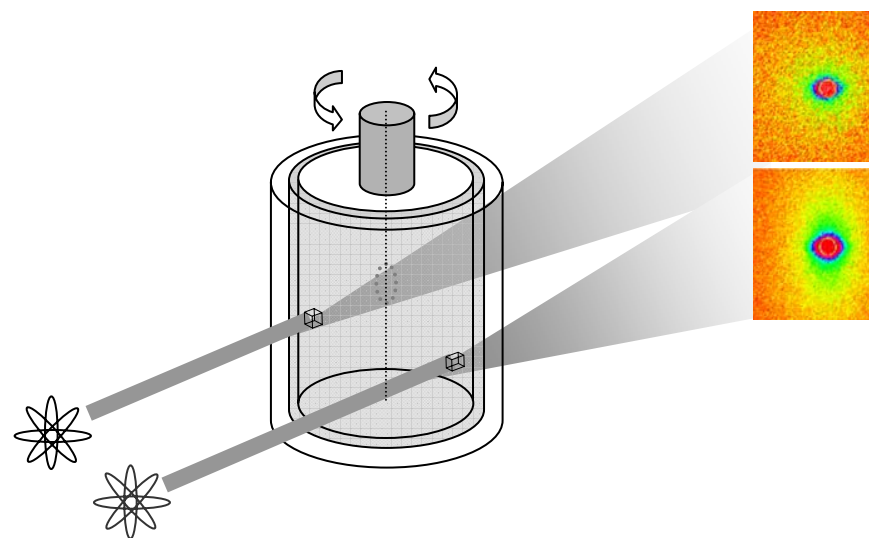
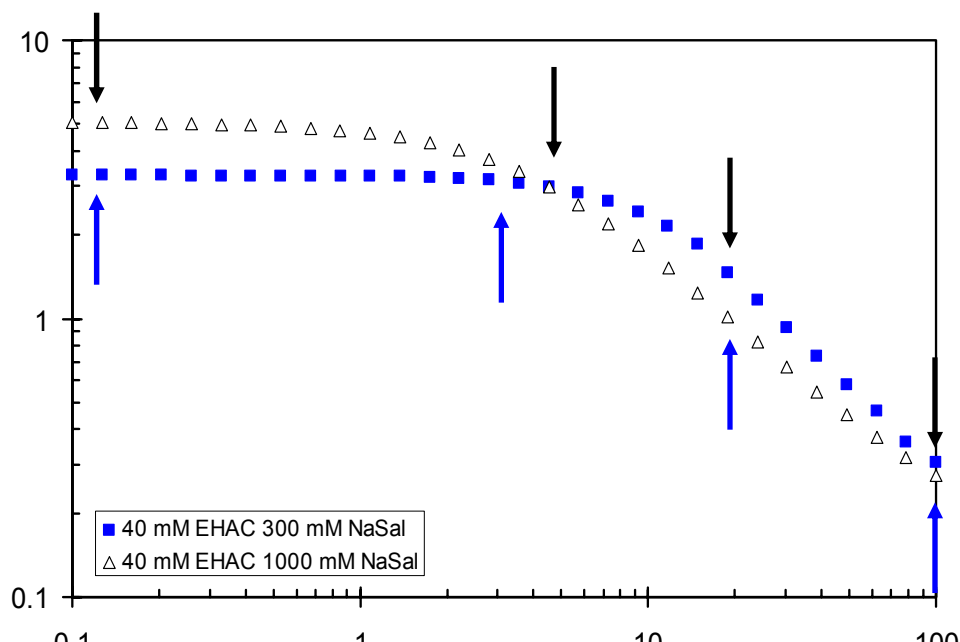
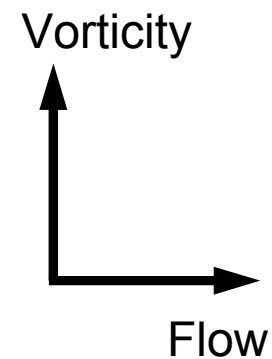
5 s<sup>-1</sup>

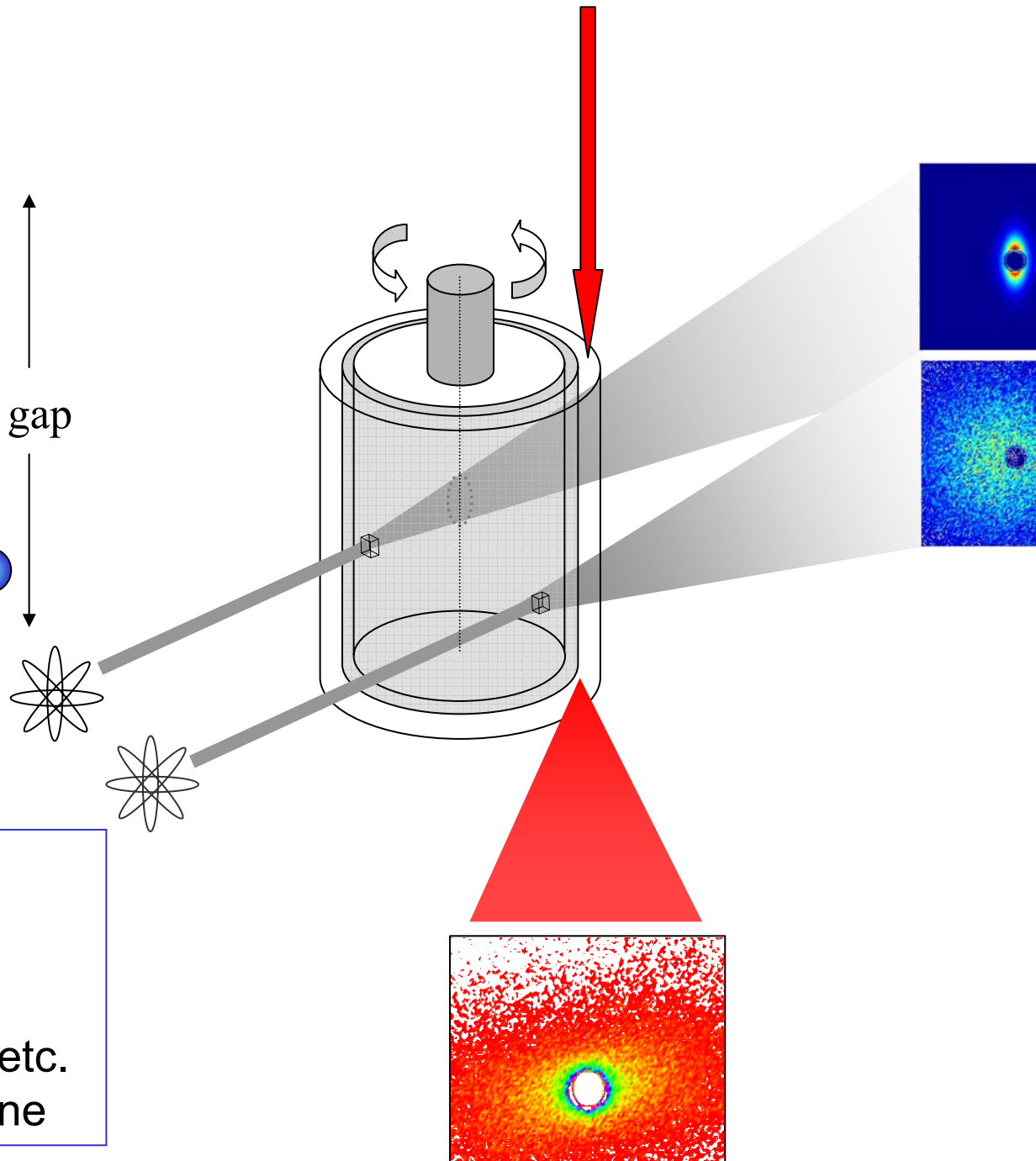
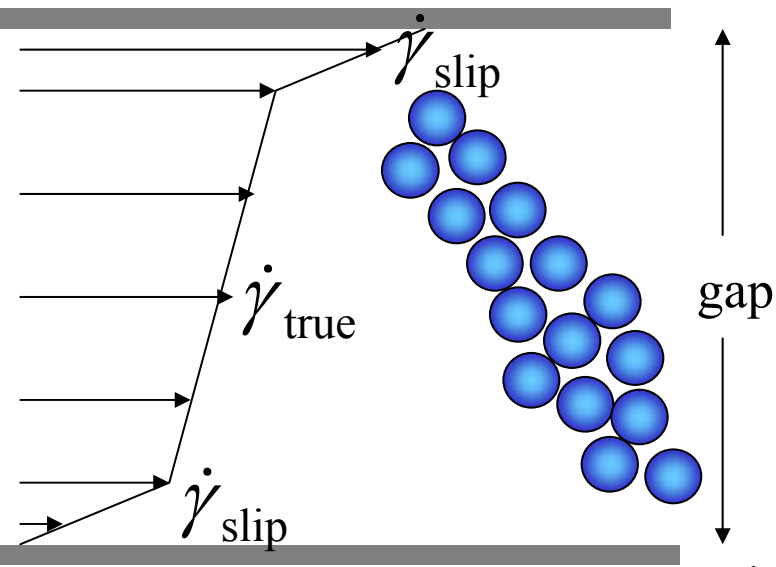


20 s<sup>-1</sup>



100 s<sup>-1</sup>





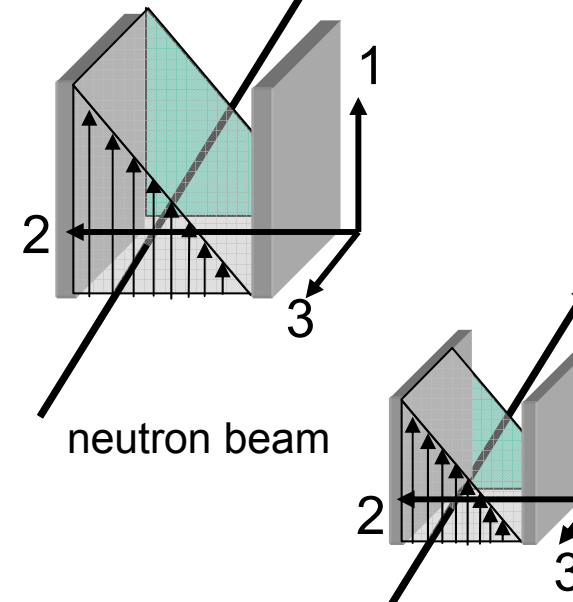
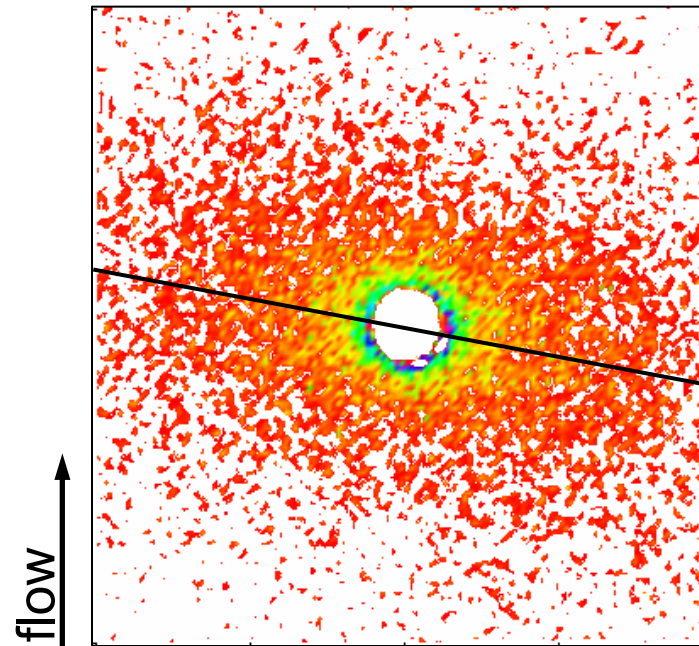
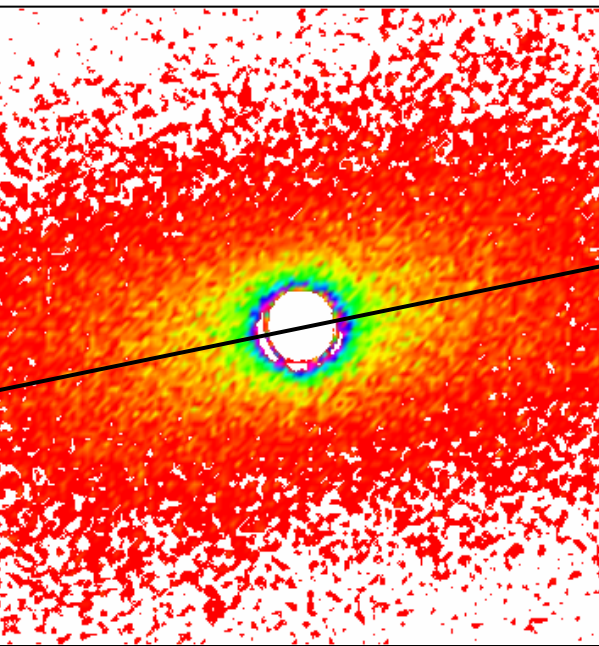
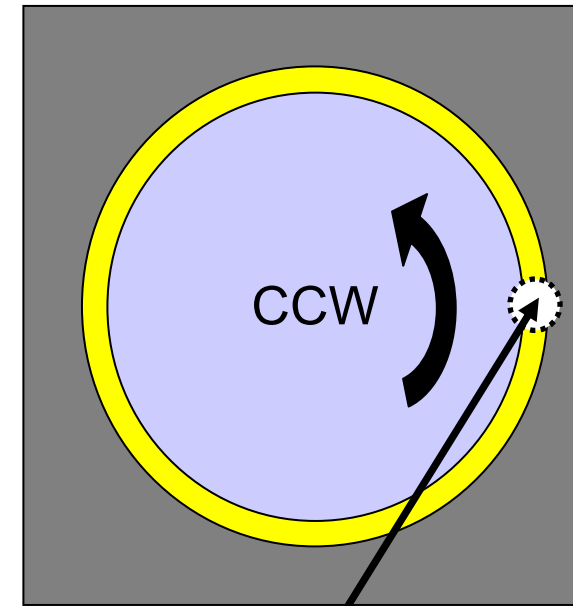
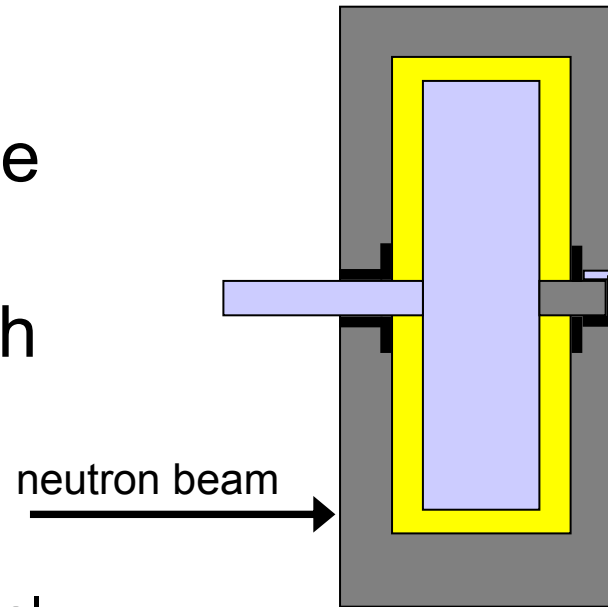
- Maximum microstructure deformation and alignment is available in the 1-2 plane
- Shear banding, wall effects, etc. are accessible only in 1-2 plane



# 1-2 Plane SANS Setup

- Aluminum Couette
- 2.0 mm gap
- 10 mm path length

40 mM EHAC 300 mM NaSal



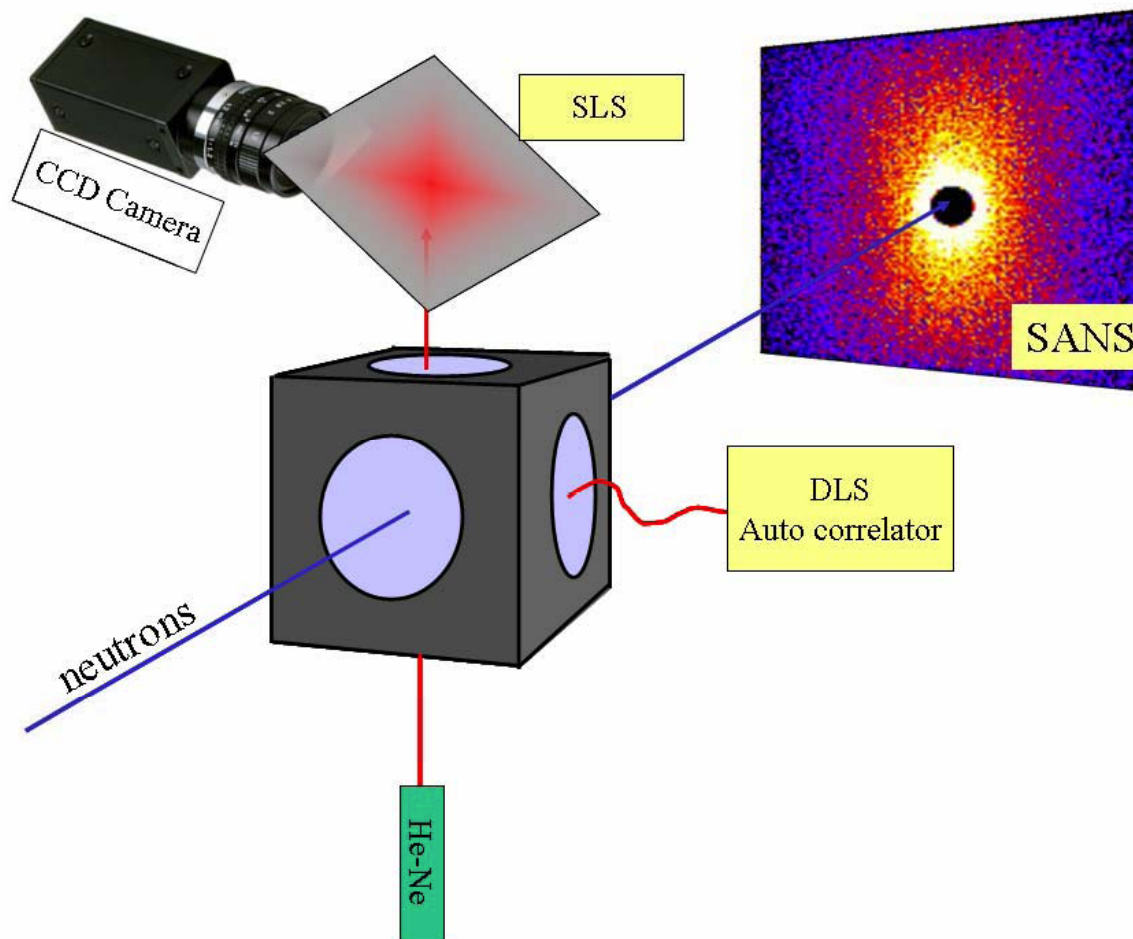
# Summary

- Dynamics of self-assembled materials under flow can be resolved

# Limitations and Challenges

- Expanded length scales
  - Lower  $q$  to see aggregation, etc.
- Dynamics needs time resolution
  - Chemical kinetics
  - Physical relaxations
- Diffusion processes (reptation!)
- External fields
  - $\underline{E}$ ,  $\underline{H}$ ,  $\underline{v}$ ,  $p$ ,  $\mu$ , ...
- Biology!



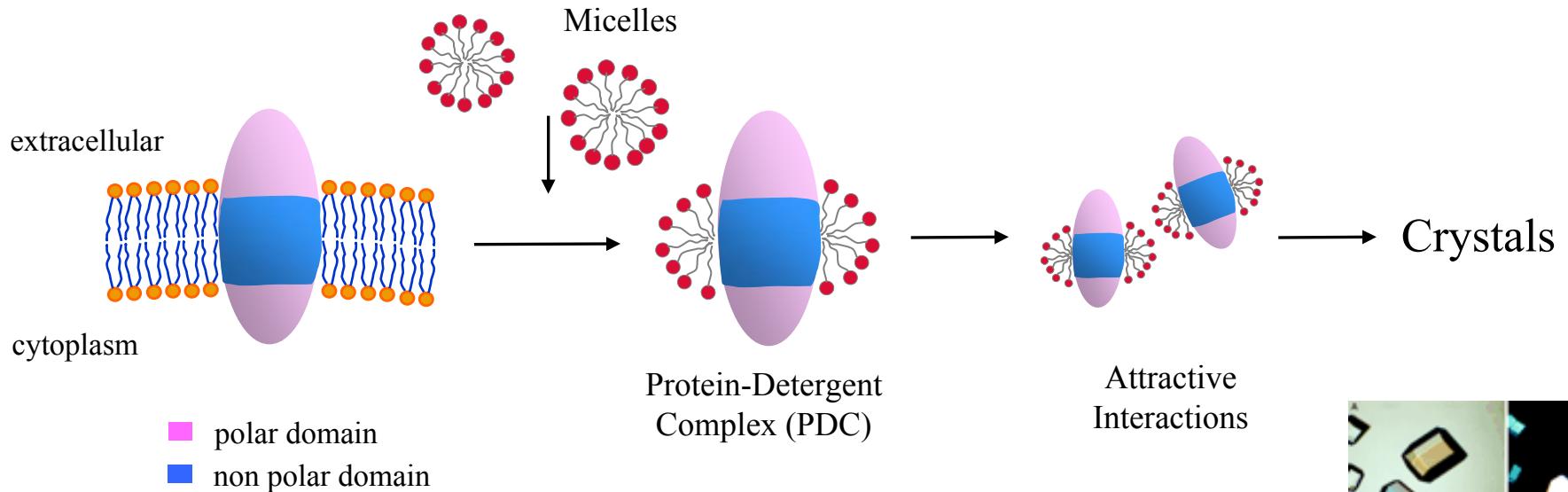


Simultaneously:

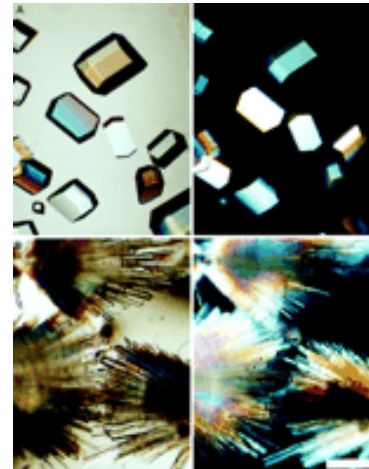
- small angle neutron scattering
- static light scattering
- fiber-optic directed dynamic light scattering

# Biological Issues

- Protein configuration and interactions with lipids in native environments



- Protein function
- Proteins in action!



# Needs

- Flux, Flux, Flux

# Useable

# Needs

- / Flux, Flux, Flux

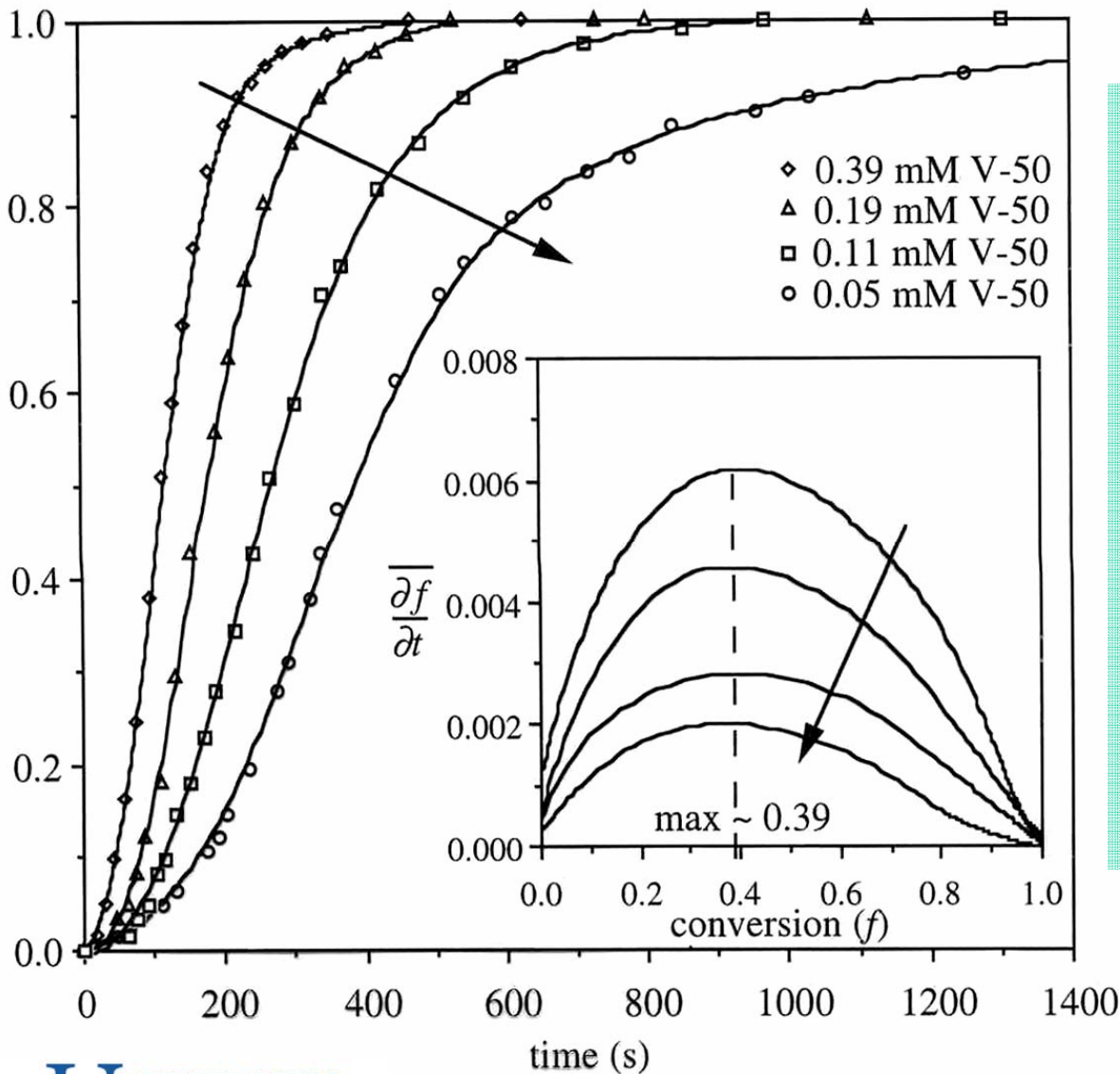
# Useable

# Needs

- / Flux, Flux, Flux
- Resolution
- But *useable* means
  - Better detectors
  - Better sample environments and sample throughput
  - Data analysis
    - Theory!

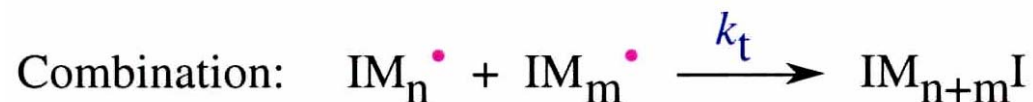
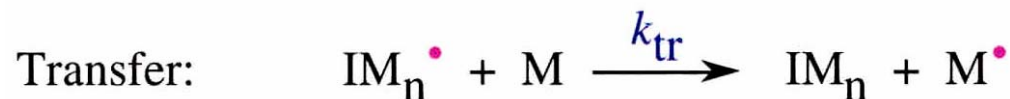
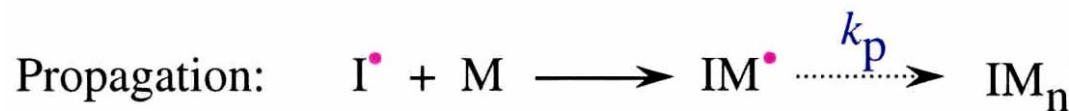
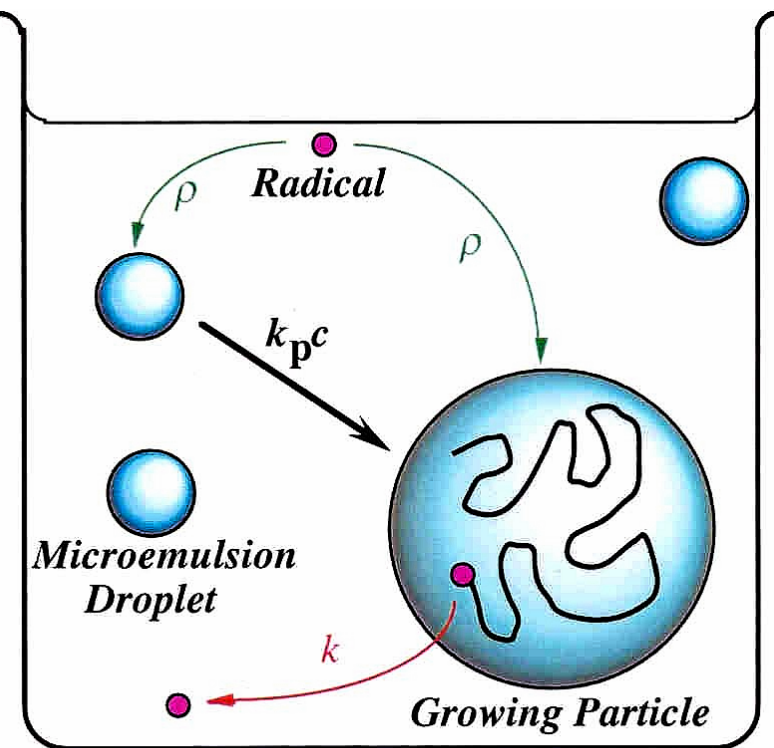


# V – 50 Polymerizations



1. Rapid polymerization
2. 100% conversions
3. Rate profile parabolic
4. Average maximum rate at about **39%** conversion

# Microemulsion Polymerization – John Morgan





# Modeling the V-50 Rate Curves

- Fundamental rate equation for addition polymerization:  
c = monomer concentration at polymerization locus  
 $[R^\bullet]$  = concentration of propagating radicals

$$-\frac{\partial c}{\partial t} = k_p [R^\bullet] c$$

- Microemulsion conversion form:

$$\frac{\partial f}{\partial t} = \frac{k_p N^*(t) c(f)}{M_o}$$

$M_o$  = monomer concentration at polymerization locus (M)

$N^*$  = propagating radical concentration in whole microemulsion (M)

# Modeling the V-50 Rate Curves (cont.)

- Assumption: Monomer concentration within polymer particles given by:

$$c = c_o (1 - f)$$

$c_o$  = initial concentration of monomer, M

- Entry rate is constant; all radicals remain active

$$N^*(t) = \rho_o t$$

$\rho_o$  = rate of radical entry, M s<sup>-1</sup>

# Modeling the V-50 Rate Curves (cont.)

- Rate Equation: 
$$\frac{\partial f}{\partial t} = \frac{k_p N^*(t) c(f)}{M_o} = \frac{k_p c_o \rho_o}{M_o} t(1-f)$$

Or

$$\frac{\partial f}{\partial t} = At(1-f)$$

Parameter:

$$A = \frac{k_p c_o \rho_o}{M_o}$$

Conversion:

$$f = 1 - \exp\left(-\frac{1}{2} At^2\right)$$

# Modeling Implications

Rate Maximum :  $\bar{f}' = \sqrt{\frac{A}{e}}$

Conversion at Maximum Rate :

$$\bar{f} = 1 - e^{-0.5} = 0.39$$

Time of Maximum Rate :  $\bar{t} = A^{-1/2}$

- Dependence on Initiator Concentration:

Assume  $\rho_o = 2k_d [I]$

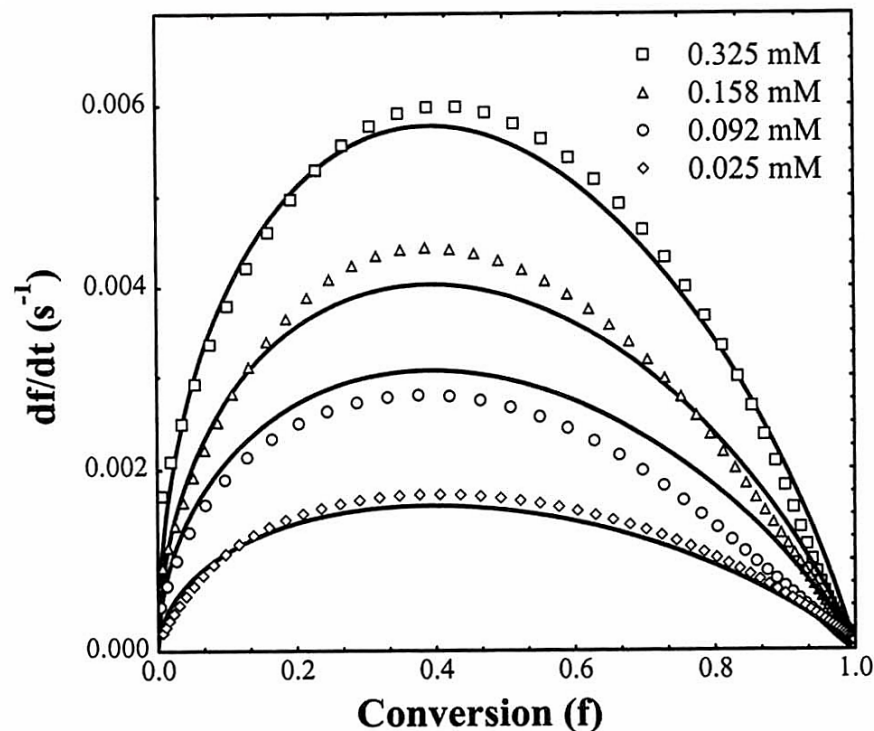
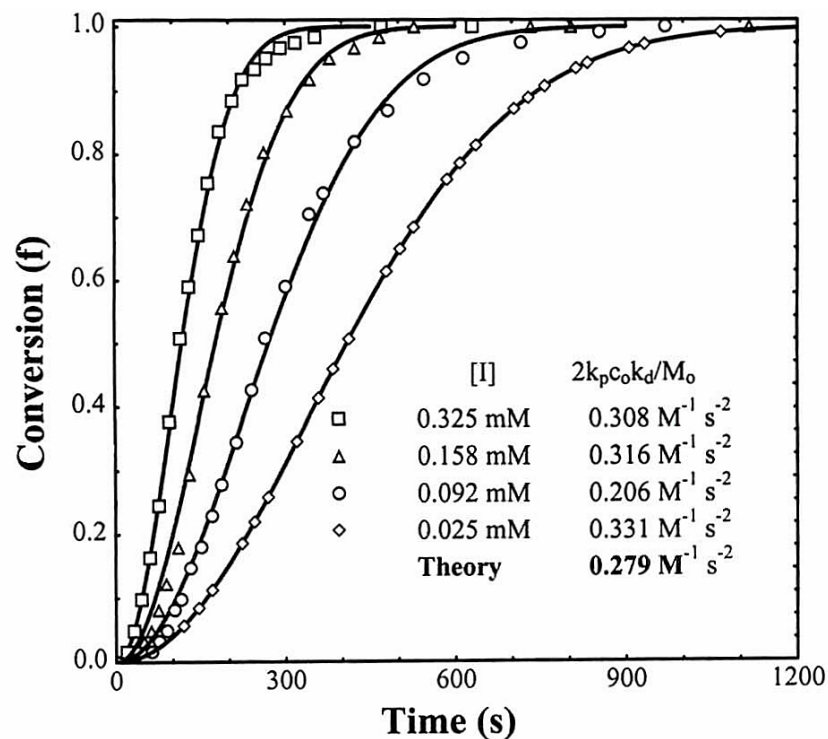
then  $A$  goes as  $[I]$

# Measured Rate Constants

- Propagation Rate Constant
  - $k_p = 995 \text{ M}^{-1}\text{s}^{-1}$  (Pulsed Laser Polymerization)
- Initiator Decomposition Rate Constant
  - $k_d = 3 \times 10^{-5} \text{ M}^{-1}\text{s}^{-1}$  (Literature)
- Initial Monomer Concentration in Droplet
  - $C_0 = 1.0 \text{ M}$  (SANS)
- Initial Monomer Concentration in microemulsion
  - $M_0 = 0.257$  (Formulation)

**N.B. No fitted parameters!**

# Comparison with Experiment



Quantitative Agreement Through to Full Conversion